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Polymersomes from amphiphilic glycopolymers containing polymeric liquid crystal grafts.

Khalid Ferji^{1,2,3}, Cécile Nouvel^{1,2}, Jérôme Babin^{1,2}, Min-Hui Li⁴, Cédric Gaillard⁵, Erwan Nicol³, Christophe Chassenieux³, Jean-Luc Six^{1,2*}

1) Université de Lorraine, Laboratoire de Chimie Physique Macromoléculaire LCPM, UMR 7375, Nancy F-54000, France

2) CNRS, Laboratoire de Chimie Physique Macromoléculaire LCPM, UMR 7375, Nancy F-54000, France

3) LUNAM Université, Université du Maine, Institut des Molécules et Matériaux du Mans UMR-CNRS 6283, Avenue Olivier Messiaen, F-72085 Le Mans cedex, France

4) Institut de Recherche de Chimie Paris, UMR8247, CNRS - Chimie ParisTech (ENSCP), 11 rue Pierre et Marie Curie, F-75231 Paris, France.

5) INRA, UR 1268 Unité Biopolymères Interactions Assemblages, F-44300 Nantes, France

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ABSTRACT: For the first time, polymersomes were obtained by self-assembly in water of amphiphilic grafted glycopolymers based on dextran polysaccharidic backbone and polymeric liquid crystal grafts (poly(diethylene glycol cholesteryl ether acrylate), PDEGCholA). After measuring the properties of these glycopolymers in term of surfactancy, the influence of their structural parameters on their self-assemblies once dispersed in water was investigated by static and dynamic light scattering and by cryogenic transmission electron microscopy (cryo-TEM). Based on the results, a proper design of Dex-g^N-PDEGCholA^F leads to hollow vesicular structure formulation known as polymersome.

Since many decades, researchers have been attempting to design model artificial membranes that can mimic the more complex biological ones by using phospholipids whose self-assembly in aqueous medium leads to vesicles commonly named liposomes.^[1] More recently, polymeric counterparts of liposomes, so-called polymersomes, have been developed through the use of copolymers.^[2] Usually, polymersomes are hollow spherical nano-objects composed of a hydrophobic compact bilayer membrane with hydrophilic inner and external surfaces. It was shown that this hydrophobic bilayer, thanks to its polymeric nature, enhances the mechanical stiffness and decreases the permeability to both solvent and solute than the ones of the liposomes.^[3-5]

Such polymersomes were often produced by the self-assembly of amphiphilic block copolymers and found applications as drug delivery systems or nanoreactors.^[6-9] Many reviews can be found dealing with the polymersome formulations from coil-coil diblock or triblock copolymers or with the experimental methods to investigate their vesicular morphology.^[10-11] In order to improve the mechanical properties of the polymeric membrane, rod-coil diblock copolymers with block containing stiff moieties such as cholesterol or polypeptide

were used. For instance, some mesogenic polymeric blocks have been considered in combination with various hydrophilic ones^[12-22] and have been shown to result in stimuli responsive polymersomes, which can be disrupted upon applying one external stimulus.^[6, 23-24] The design of smart polymersomes either responding to pH, temperature, redox, magnetic field or light stimuli^[6, 11, 25-27] has also been achieved thanks to the processing of new polymerization techniques and of chemical couplings. Biological polymers^[7-9, 28-31] like polypeptides, polynucleotides or polysaccharides as well as biodegradable polymers such as polylactic acid (PLA) or poly(ϵ -caprolactone) (PCL) were also considered to produce polymersomes of interest for biological applications, especially when combined with biocompatible polyethylene glycol (PEG) block that leads to stealthy polymersomes.^[32] However, the formation of polymersomes using other polymeric architectures than block copolymers such as graft copolymers has been reported in a far less extent. To the best of our knowledge, only four graft copolymers forming polymersomes have been investigated in the literature so far: poly(2-hydroxyethyl aspartamide)-g-oligo(lactic acid)^[33], poly(b-amino ester)-g-(PEG/PLA)^[34], chitooligosaccharide-g-

PCL^[35] and poly(3-hydroxyoctanoate-co-3-hydroxyundecanoate)-g-PEG^[36].

The aim of the present study was to investigate the self-assembling propensity in solution and at liquid/liquid interface of graft glycopolymers (called Dex-g^N-PDEGCholA^F) containing liquid-crystal (rod) grafts that were previously reported by us.^[37] In these glycopolymers, Dextran is used as the hydrophilic polysaccharidic backbone and poly(diethylene glycol cholesteryl ether acrylate) (PDEGCholA) as hydrophobic polymeric grafts. Various Dex-g^N-PDEGCholA^F were synthesized using a "grafting from" four-step strategy described in scheme S1.^[37] N and F, which are respectively the number of PDEGCholA grafts per 100 glucopyranosic units of dextran backbone

and the weight fraction of PDEGCholA in glycopolymer, have been varied in some extent (Table 1).^[37]

In bulk, PDEGCholA grafts appeared to exhibit both single glass transition and liquid crystal to isotropic transition temperatures. Between these two transition regimes, the bulk displayed at the microscopic scale typical smooth fan-shape birefringent textures, which indicated a smectic A phase further confirmed by X-ray scattering measurements.^[37] By comparing the lamellar period to the PDEGCholA monomer length, we established that an interdigitated smecticA phase (SmA_d) was formed. This peculiar mesomorphism of the grafts was preserved once they were grafted onto a dextran backbone.^[37]

Table 1. Characteristics of Dex-g^N-PDEGCholA^F and of their corresponding nanostructures formed in aqueous media.

Entry	Copolymers characteristics			Solvent (before dialysis)	Nanostructure characteristics				
	Dex-g ^N -PADEGChol ^{F a)}	\overline{M}_n of grafts [g/mol] ^{b)}	\overline{D} ^{c)}		R_H ^{d)} [nm]	R_G ^{e)} [nm]	$\frac{R_G}{R_H}$	Polydispersity ^{f)}	t ^{g)} [nm]
C1	Dex-g ²⁴ -PADEGChol ⁹⁰	5.800	1.2	DMSO	-	-	-	-	-
				THF	-	-	-	-	-
C2	Dex-g ^{6,6} -PADEGChol ⁷³	6.800	1.2	DMSO	-	-	-	-	-
				THF	-	-	-	-	-
C3	Dex-g ⁵ -PADEGChol ⁷⁹	12.600	1.2	DMSO	50	54	1.1	1.2	18.5 ± 1.0
				THF	68	63	0.9	1.1	20.0 ± 1.0*
C4	Dex-g ^{1,5} -PADEGChol ⁵⁰	11.000	1.3	DMSO	48	47	1.0	1.3	3.5 ± 0.5
				THF	62	67	1.1	1.2	6.5 ± 0.5

a) N and F are the number of PDEGCholA grafts per 100 glucosidic units of dextran backbone and the weight fraction of PDEGCholA in amphiphilic glycopolymer, respectively. \overline{M}_n and \overline{D} of dextran were estimated equal to 34800 g mol⁻¹ and 1.14, respectively [37].

b) \overline{M}_n of each PDEGCholA graft estimated from $\frac{[DEGCholA]_0}{[Br]_0} \times M_0 \times \text{conv}$, where $[DEGCholA]_0$ and $[Br]_0$ are the initial molar concentrations of monomer and ATRP initiator groups carried by the macroinitiator (after steps i and ii, Scheme S1, [37]), respectively. M_0 is the molecular weight of monomer unit of PDEGCholA (528 g mol⁻¹).

c) Dispersity of the corresponding acetylated copolymers (before step iv, Scheme S1, [37]) measured by SEC in THF.

d) Hydrodynamic radius determined by dynamic light scattering.

e) Radius of gyration determined using Guinier plot.

f) See supporting information.

g) Membrane thickness of vesicles determined by fitting the Rayleigh ratio of the suspensions using equation S4.

*: Membrane thickness was estimated equal to 25 nm from Figure 2a.

In the present paper, we will focus more on the properties displayed by these graft glycopolymers when dispersed in solution. First, we will establish that these graft copolymers exhibit interfacial activities. Then, we will focus on the influence of the chemical structure of these copolymers onto the characteristics of their self-assemblies in water. Finally, we will show that these self-assemblies are actually of polymersome type.

Results and Discussion

1. Interfacial properties of Dex-g^N-PDEGCholA^F glycopolymers

In order to estimate the propensity of such glycopolymers to stabilize interfaces, we have dissolved two of them (entries C₃ and C₄, Table 1) in toluene (concentration lower than 0.2 g L⁻¹). After waiting 12 hours at room temperature, in order to reach thermodynamic equilibrium, the interfacial properties of each Dex-g^N-PDEGCholA^F were evaluated by steady state interfacial tension (γ) measurements at the toluene/water interface as function of copolymer concentration. As shown in Figure S1, the interfacial tension of a pure toluene/water interface was equal to 28.8 mN m⁻¹. By adding the graft glycopolymer, γ steadily decreased until A₁ = 1.5 mg L⁻¹ (case of Dex-g²-PDEGCholA⁷⁹, entry C₃, Table 1) or A₂ = 6.2 mg L⁻¹ (case of Dex-g¹⁵-PDEGCholA⁵⁰, entry C₄, Table 1) where the decrease of γ was much more pronounced. As shown in Table 1, each studied glycopolymer exhibits $\bar{D} \neq 1$ meaning they are heterogeneous in composition (number and length of graft). Nevertheless, as the length of the PDEGCholA grafts is roughly the same for both copolymers, the number of grafts onto the dextran backbone could account for such a difference: decreasing this number increased the concentration at which γ decreased. Above A₁ or A₂, γ decreased more steeply with increasing polymer concentration. Unfortunately, when the Dex-g^N-PDEGCholA^F concentration was too high, we observed a white polymeric layer located at the interface that prevented reliable measurements of γ . Consequently, the concentration range where the investigations were possible is rather limited. Nevertheless, the decrease of γ upon increasing polymer concentration can be related to their ability to stabilize toluene/water interface as molecular surfactant does. Furthermore the grafted nature of these amphiphilic copolymers and mesomorphic transitions of grafts could account for the broadening of the transition, which is far from being critical by opposition to what is encountered with molecular surfactants or even amphiphilic diblock copolymers.

2. Self-assembly of Dex-g^N-PDEGCholA^F glycopolymers

The ability for self-assembly of these graft glycopolymers was further investigated in water (figure S2). First, Dex-g^N-PDEGCholA^F were dispersed either in THF at 25°C or in DMSO at 100 °C, and then dialyzed against water (a selective solvent for dextran) during two days. Finally, the morphology of their self-assemblies was investigated by dynamic and static light scattering (DLS and SLS, respectively) and by cryogenic transmission electron microscopy

(cryo-TEM). The characteristics of these nano-objects are summarized in table 1. Nanostructures obtained with the copolymer Dex-g¹⁵-PADEGChol⁵⁰ (entry C₄, table 1) dissolved in DMSO or in THF are denominated C₄^{DMSO} or C₄^{THF}, respectively.

Depending on the number and the length of the grafts, two situations have been encountered. For copolymers containing a high number of short hydrophobic grafts (\bar{M}_n of graft < 7.000 g mol⁻¹, entries C₁ and C₂, Table 1), a precipitate was formed during dialysis, whatever the organic solvent we used (THF or DMSO). By opposition, when copolymers containing a moderated number of long grafts (\bar{M}_n of graft # 11.500 g mol⁻¹, entries C₃ and C₄, Table 1) were considered, nano-objects were produced.

In these last cases, the dispersions are characterized by a single distribution of relaxation times in DLS over the whole scattering range investigated (from $q=4.6 \times 10^{-3} \text{ nm}^{-1}$ to $2.55 \times 10^{-2} \text{ nm}^{-1}$) as shown in figure 1 that displays typical examples of autocorrelation functions measured at the same scattering angle for each dispersion. The relaxation time derived from the auto-correlation functions displayed a q^2 dependence (figure S3), which meant that diffusive motions of the scatterers were probed and which allowed us to estimate their hydrodynamic radius (R_H) using the Stokes-Einstein equation (Equation S2). As shown in table 1, R_H ranged between 47 nm and 68 nm with a polydispersity close to 1 (see supporting information). In addition, the radius of gyration (R_G) for each dispersion was estimated by static light scattering measurements using the Guinier approximation. As shown in table 1, R_G ranged between 47 nm and 63 nm. For all the nano-objets (C₃^{DMSO}, C₃^{THF}, C₄^{DMSO} and C₄^{THF}), the ratio R_G/R_H was estimated to be close to 1, which accounted for a vesicular morphology of the self-assemblies^[38, 39] and

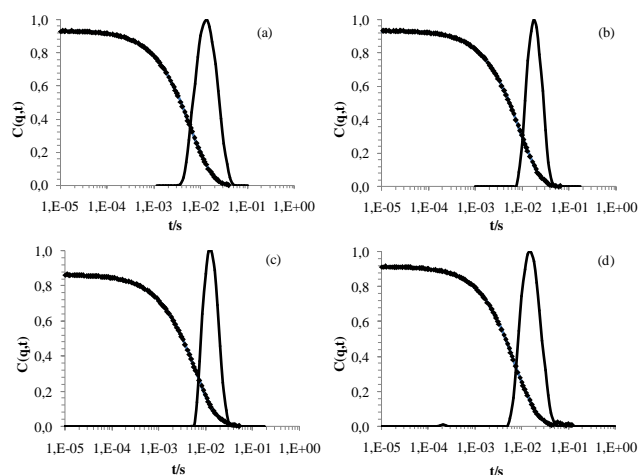


Figure 1. Autocorrelation functions $C(q,t)$ as a function of time and corresponding relative distribution of relaxation times recorded at a scattering wave vector $q=4.6 \times 10^{-3} \text{ nm}^{-1}$ ($\theta=20^\circ$) for different nanostructures dispersed in water. (a) C₃^{DMSO}, (b) C₃^{THF}, (c) C₄^{DMSO}, (d) C₄^{THF}.

indicated that hollow spherical structures were obtained whatever the initial non selective solvent used (THF or DMSO).

To further investigate the morphology of the nanostructures, we have studied the dispersions of C_3 and C_4 by cryogenic Transmission Electronic Microscopy (cryo-TEM). As shown in Figure 2a (case of C_3^{DMSO}), vesicles with radius equal to 35 nm (z-average) were actually observed. A schematic representation of such a vesicle is drawn in figure 2b. The higher radius of vesicles found by DLS analysis compared to that estimated by cryo-TEM could be explained by hydration effects of the dextran corona that is extended in water. Moreover, dextran is less visible by cryo-TEM considering its swelling in water, leading to a low electronic contrast compared to water.^[40] Supplementary images of the same nano-objects yet observed by TEM are presented in supporting information (figure S4).

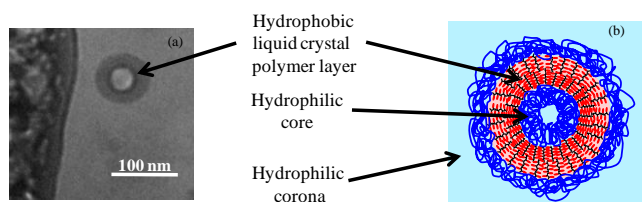


Figure 2. (a) Cryo-TEM image of polymersome obtained in water after dialysis of Dex-g⁵-PADEGChol⁷⁹ dispersed in THF(case of C_3^{THF}). (b) schematic representation of such vesicle.

A more quantitative description of the self-assemblies has been achieved thanks to Static Light Scattering measurements. Figure 3 displays the scattered intensity as function of the scattering vector for each suspension (C_3^{DMSO} , C_3^{THF} , C_4^{DMSO} and C_4^{THF}). The results could be fitted with a model developed for vesicles (Equation S4).^[41, 42] We can then derive the thickness of the membrane (t) knowing the hydrodynamic radius of vesicle, the refractive index of aqueous medium (1.33) and that of the membrane made of PADEGChol (1.38) as reported in table 1. A higher membrane thickness is obtained in case of copolymer Dex-g⁵-PADEGChol⁷⁹, which contains a higher number of grafts compared to that obtained for Dex-g¹⁵-PADEGChol⁵⁰. That could be explained by the stiff nature of the PADEGChol grafts.^[37] Actually, in case of C_3^{DMSO} and C_3^{THF} suspensions, the polymeric membrane may consist in partially interdigitated grafts due to a higher grafting density onto the dextran backbone leading to steric hindrance. By opposition, in case of C_4^{DMSO} and C_4^{THF} , the polymeric membrane may consist in fully interdigitated grafts as shown in the schematic representation of figure 3. The difference in the membrane thickness for C_4^{DMSO} and C_4^{THF} samples could be related to the better solubility of PDEGChol grafts in THF. Improved studies will be done in the future to confirm these observations.

Conclusions

After the synthesis of this first class of grafted glycopolymer containing liquid-crystal PDEGCholA grafts and dextran backbone and the study of their mesomorphic properties, the ability of Dex-g^N-PDEGCholA^F to stabilize liquid-liquid interfaces were firstly proved. Then, self-assembly of such glycopolymer was studied after dissolution in THF or DMSO and dialysis against water. Dynamic and static light scattering allowed the observation of spherical nanostructures (R_H between 47 and 68 nm, R_G/R_H close to 1), which are formed depending on the Dex-g^N-PDEGCholA^F structure. Finally, cryo-TEM observations clearly showed an hollow vesicular structure known as polymersome. Based on these results, a proper design of Dex-g^N-PDEGCholA^F led to polymersome formulations. It seems that moderate number of long PDEGCholA grafts allows an adapted copolymer folding and a bilayer membrane formation, which is stabilized due to the liquid-crystal properties of PDEGCholA grafts. Deeper analyses of such Dex-g^N-PDEGCholA^F self-assemblies will be performed to gain a better understanding of the organic solvent influence and to investigate in-depth these vesicular morphologies. Nevertheless, such graft copolymers could have an interest for drug delivery systems: firstly, the materials involved in these copolymers (dextran and PDEGCholA) are biocompatible. Secondly, neutral dextran backbone may confer a stealthy character to nanostructures like PEGylation coverage.^[43] Thirdly, hydroxyl groups present on dextran could be used for functionalizing the surface of the nano-objects with sufficient and consequent amount of "bioactive" molecules like specific peptides for instance.^[43]

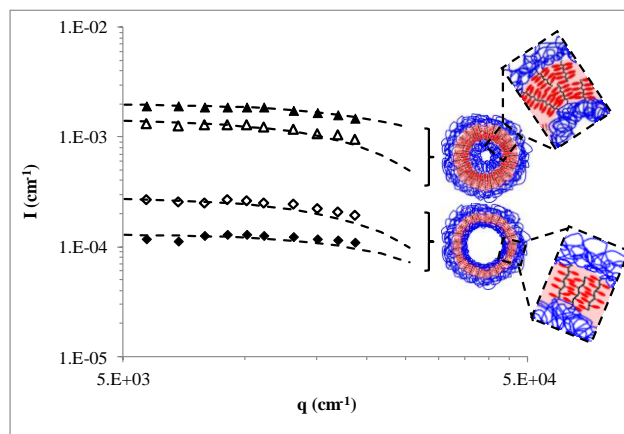


Figure 3. Scattered intensity (I) of vesicles as function of scattering vector (q) of: (\blacktriangle) C_3^{DMSO} , (\triangle) C_3^{THF} , (\blacklozenge) C_4^{DMSO} and (\lozenge) C_4^{THF} . Dashed lines are fits to equation S4. Schematic representations of vesicles and of their membranes are given.

ASSOCIATED CONTENT

Supporting Information: Copolymers synthesis strategy, Interfacial Tension Measurements, Copolymers self-assembly procedure, Dynamic and static light scattering (SLS and DLS), Transmission Electronic Microscopy.

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AUTHOR INFORMATION

Corresponding Author

* jean-luc.six@univ-lorraine.fr

Author Contributions

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